

Structural Characterization of Borate Glasses Containing Zinc and Manganese Oxides

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Abstract

We have investigated the effect of inclusion of two transition metal ions (TMI) on structure and optical properties of borate glass system having composition $x\text{MnO}_2 - y\text{ZnO} - (100 - x - y) \text{B}_2\text{O}_3$ ($9 \leq x \leq 12$, $36 \leq y \leq 48$) prepared by melt quenched route. Thermal study by using a differential scanning calorimeter (DSC) reveals that the glass transition temperature (T_g) and crystallization temperature (T_c) of the glasses increases with the increase of borate content in the system. Fourier transform infrared (FTIR) spectra indicate that inclusion of TMI produces BO_3 and BO_4 structural units by breaking the boroxol (B_3O_6) ring. The optical band gap energy estimated from ultraviolet-visible spectra shows a decreasing tendency when TMI are incorporated in the borate structure.

Keywords: Borate Glass, Differential Thermal Analysis (DTA), Infrared (IR) Spectroscopy, Optical Band Gap

1. Introduction

Glasses are receiving considerable attention due to their unique properties like hardness, good strength, transparency and excellent corrosion resistance. X-ray diffraction (XRD), infra-red spectroscopy (IR), differential scanning calorimetry (DSC) studies has been extensively employed over the years to investigate the structure of glasses [1-4]. Borate glasses, in particular, have been the subject of numerous infra-red studies due to their structural peculiarities [5-8]. In pure B_2O_3 glass structure most of the boron is involved in B_3O_6 (boroxol) ring. Addition of modifier breaks boroxol ring and thereby produced BO_3 and BO_4 units [6,8]. In addition, modifier also changes the physical properties along with structural modifications.

Recently, the study of oxide glasses doped with transition metal ions (TMI) has received considerable attention due to their attractive combination of physical and chemical properties. TMI doped borate glasses have application in microelectronics, optical glasses and solid state laser [9-11]. Continued effort for the development of new glassy materials either by doping or by adding TMI, and the study of their novel properties is highly

relevant due to their potential applications in various technological fields [12,13]. Keeping in mind the very fact of creating novel functionalities we have chosen an uncommon glass system. We report here the preparation, structural characterization and optical properties of manganese and zinc oxide containing B_2O_3 glass with an intention to precipitate Mn-doped ZnO crystal in the borate glass matrix, which may lead to a new composite spintronics material.

2. Experimental Procedure

2.1. Preparation

Multicomponent transition metal oxide glasses containing MnO_2 - ZnO - B_2O_3 having different compositions, presented in **Table 1**, have been prepared from analytical grade precursors MnO_2 , ZnO and B_2O_3 . Batches of 5 gm sample were prepared by taking weighted amounts of three oxides in an alumina crucible and melting mixture in a precisely controlled high temperature furnace (Thermolyne type 46100) at a temperature 1200°C in ordinary air atmosphere. Melted mixtures were repeatedly swirled to ensure complete homogenization. The

Table 1. Nominal chemical composition of the glasses (mol%).

| MnO ₂ | ZnO | B ₂ O ₃ |
|------------------|-----|-------------------------------|
| 9 | 36 | 55 |
| 10 | 40 | 50 |
| 11 | 44 | 45 |
| 12 | 48 | 40 |

glasses were prepared by quenching the melted materials between two brass plates.

2.2. Characterization

X-ray diffraction technique was used to check for possible crystallinity of the samples after quenching and annealing. We have utilized Philips (Philips PW 1050/51) x-ray powder diffractometer with CuK α radiation. Differential scanning calorimetry study was carried out on a Shimadzu DSC-60 in the temperature range 30°C - 600°C at a constant rate 10°C/min. under ordinary air atmosphere using aluminum pan. The accuracy in determining the T_g and T_c were $\pm 3^\circ\text{C}$. Scanning electron microscopy (SEM) study was performed in a Hitachi made instrument (S-3000N). Vibrational spectra of various as-prepared glasses were obtained by KBr pellet technique in the range 400 - 4000 cm⁻¹ using a Nicolet Magma-IR (750, Series II) spectrophotometer. The optical absorption spectra of as-prepared glass samples were recorded at room temperature in the range 200 - 800 nm using a double beam Hitachi spectrophotometer (model U3410).

3. Result and Discussions

3.1. XRD and SEM Study

Figure 1 presents the XRD pattern of the sample containing 45% B₂O₃ which is typical for other samples. XRD patterns of all the as-prepared samples show no sharp Bragg's peak, but only a broad diffuse hump around low angle region. This is the clear indication of amorphous nature within the resolution limit of XRD instrument. Scanning electron microscopy studies of these as-prepared samples also exhibit a clear surface without the presence of any microstructure. **Figure 2** shows a typical SEM micrograph of as-prepared sample having 55% B₂O₃. Absence of microstructure in SEM picture also indicates the amorphous as well as homogeneous nature of the samples.

3.2. DSC Analysis

Figure 3 shows the DSC curve of as-prepared sample

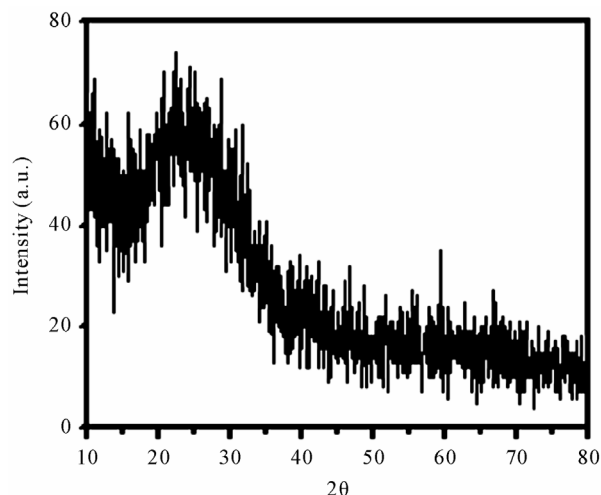


Figure 1. XRD patterns of the sample containing 45 mole% B₂O₃.



Figure 2. SEM micrograph of as-prepared sample having 55 mole% B₂O₃.

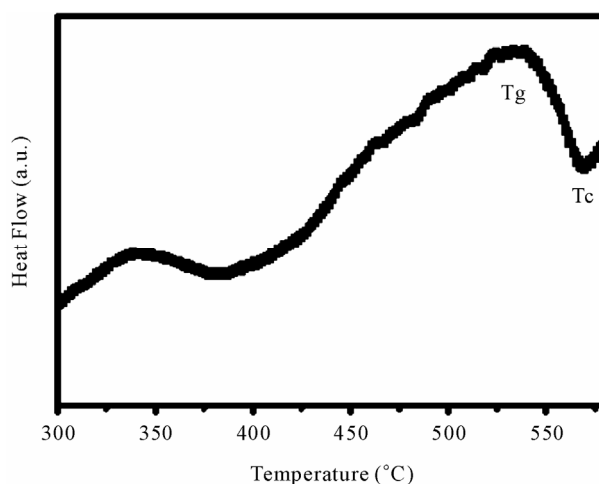


Figure 3. Differential scanning calorimetric curve of 40 mol% B₂O₃ glass.

having 40 mol% B₂O₃. This curve clearly shows one endothermic peak and one exothermic peak.

The endothermic peak corresponds to the glass transition while the exothermic peak indicates the crystallization point of the glass. The glass transition (T_g) as well as crystallization temperatures (T_c) are estimated by the slope intercept method. The nature of the DSC curves is typical for other glass compositions.

Thermal study of the glasses were performed because any change in the coordination number of network forming atoms, or the formation of nonbridging oxygen, is known to be reflected in the T_g. **Figure 4** illustrates the variation of T_g and T_c with compositions. DSC study reveals that both T_g and T_c increase monotonically with the increase of B₂O₃ content, which is the network former here. It is reported that generally T_g and T_c increases with the increase of network former/glass former [14] which is observed in this present study also. However several reports of germanate anomaly are there which shows a decrease in T_g with the increase of GeO₂ content in the glass system [15]. A maximum in the T_g vs. B₂O₃ content curve is also reported [16]. It is believed that T_g is depend on the strength of chemical bonds in the structure. TMI in general, plays the role of a network modifier and non-bridging oxygen increases with the increase of TMI content in the glass system. Increase of non-bridging oxygen indicates the breaking of chemical bonds, which in turn decrease the T_g. Increase of T_g with the increase of network former, in other words, decrease of TMI, indicates the increase of strength and connectivity of the glass structure in this case.

3.3. FTIR Study

Infrared spectroscopy has proved to be an important tool for the investigation of structure and dynamics of disorder materials. IR spectra of materials may help to get the idea of the nature of vibration in a disorder system [16]. The room temperature vibration spectra of the glasses were obtained using KBr pellet technique in the range 400 - 4000 cm⁻¹. A typical FTIR spectrum of the as-prepared glass containing 50% B₂O₃ is shown in **Figure 5**. As expected, these spectra exhibit broad absorption bands as a consequence of the general disorder in the network, mainly due to a wide distribution of structural units occurring in these glasses.

The band marked as A attributed to the presence of transition metal ions in bi-valent state (Zn²⁺, Mn²⁺). The absorption bands marked as B, C and D are due to borate matrix. Details of the appeared peaks are presented in **Table 2**. The peak assignment is consistent with other published work [17-19]. Absence of peak around 806 cm⁻¹, which is clear from the inset of **Figure 5**, indicates

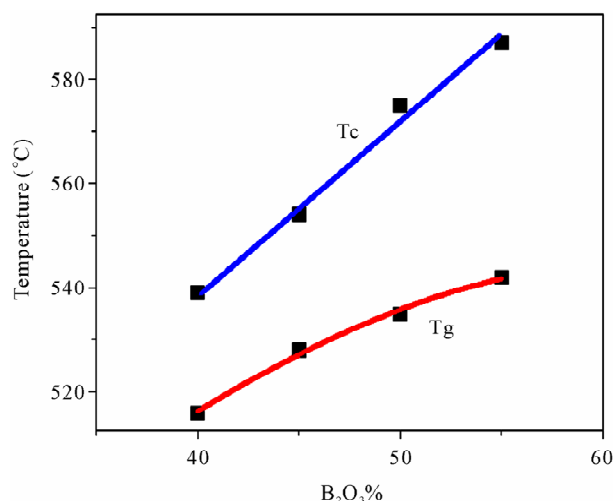


Figure 4. Compositional dependency of glass transition and crystallization temperature (Line connecting the data points is a guide for the eye).

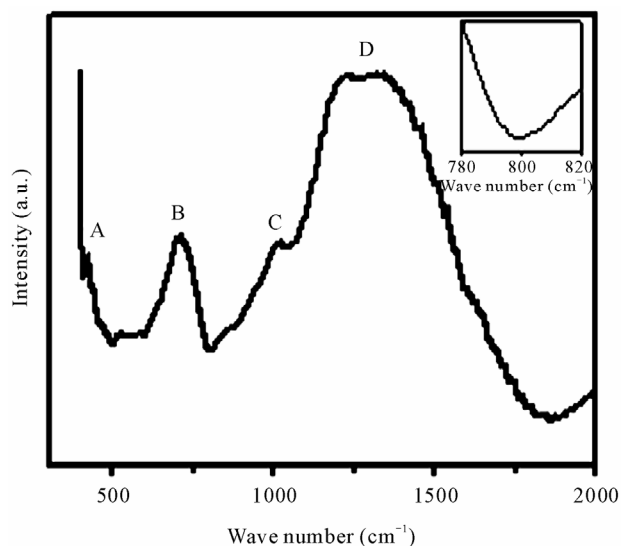


Figure 5. Room temperature FTIR spectra of zinc manganese borate glasses for 50 mol% B₂O₃. Inset: Magnified version of FTIR curves to prove the absence of boroxol rings.

Table 2. Various absorption peak positions obtained from FTIR spectra.

| Band | Position of Band (cm ⁻¹) | Assignment |
|------|--------------------------------------|---|
| A | ~425 | Vibration of metal cations such as Zn ²⁺ /Mn ²⁺ |
| B | ~700 | Bending of B-O-B linkage |
| C | ~1020 | B-O stretching of BO ₄ tetrahedra |
| D | ~1280 | Asymmetric stretching of B-O of trigonal BO ₃ |

that borate network does not contain any boroxol ring [17]. The broadness of D band is due to the presence of Zn^{2+} in the system. Generally, in pure B_2O_3 glass most of the boron is involved in B_3O_6 boroxol rings [18]. The addition of TMI breaks these rings and increasingly BO_3 and BO_4 units are formed which is, reflected in our samples also [20]. No detectable variation in peak positions and band shape are observed with the change of composition.

3.4. Optical Absorption Study

The fundamental optical band gap of the glasses has been computed based on their UV-Vis absorption spectra, for understanding their optically induced transitions. There are two types of optical transition, which can occur at the fundamental absorption edge of crystalline and noncrystalline materials. They are direct and indirect transitions. In both the cases, electromagnetic waves interact with the electrons in the valence band, which are raised across the fundamental band gap to the conduction band. For photon energies $h\nu$ just above the fundamental edge, the absorption α follows the standard relation,

$$\alpha = A(h\nu - E_g)^{1/2} / h\nu$$

where A is a constant and E_g is defined as the energy band gap. A typical plot of $(\alpha h\nu)^2$ versus $h\nu$ for the glass having 55% B_2O_3 is presented in **Figure 6**. Extrapolation of this plot to $\alpha^2 = 0$ gives the optical band gap E_g for direct transition. We have estimated the optical band gap for all the glasses, which varies from 3.45 - 4.05 eV for direct transition while B_2O_3 content changes from 40% - 55%. The value of E_g for indirect transition is obtained by extrapolation of $(\alpha h\nu)^{1/2}$ versus $h\nu$ plot to $\alpha^{1/2} = 0$ [21]. The optical band for indirect transition varies from 3.39 - 3.60 eV while B_2O_3 content changes from 40% - 55%. **Figure 7** reveals the variation of optical bandgap with the change of composition. It can be observed from **Figure 7** that both the direct and indirect band gap increases with the increase of B_2O_3 . In other words, the optical bandgap decreases with the increase of TMI concentration. This can be attributed to the structural changes that are taking place with the introduction of TMI. Inclusion of TMI in borate structure may create some defect states in the midgap, which is responsible for the decrease of bandgap. The broadness of the absorption edge may be due to the presence of localized state in the band tails.

This paper is a part of more general study concerning the synthesis, structural characterization and optical study of borate glass system. It is noteworthy to mention here that we have annealed one glass sample at 500°C for 60 min. XRD pattern of annealed sample confirms the

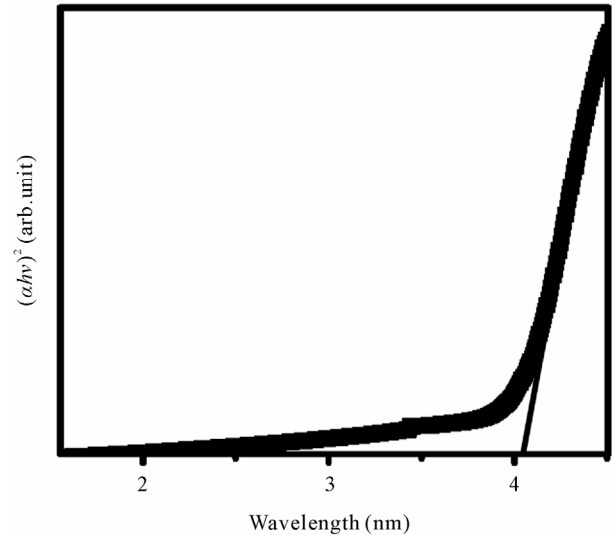


Figure 6. $(\alpha h\nu)^2$ versus $h\nu$ plot for the glass containing 55 mol% B_2O_3 .

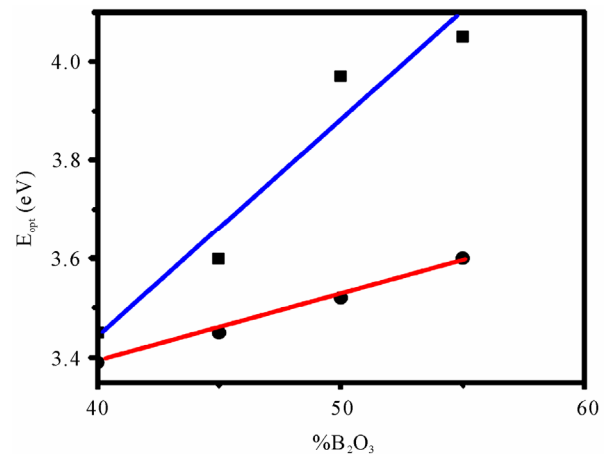


Figure 7. Compositional dependency of optical bandgap Blue line (■): Direct transition, Red line (●): Indirect transition (Line connecting the data points is a guide for the eye).

precipitation of nanocrystalline Mn-doped ZnO, which also shows magnetization at room temperature. This result indicates the possibility towards the development of borate based spintronic materials. Details of the magnetization are reported elsewhere [22].

4. Conclusions

In conclusion, we are able to prepare series of uncommon manganese zinc oxide containing B_2O_3 glasses through melt-quenched technique and report the experimental investigation of thermal stability and spectroscopic studies. Both the x-ray diffraction and SEM studies confirm the amorphous nature of the as-prepared

glasses. The effect of TMI doping on structure and optical property is discussed. In addition, we have demonstrate that magnetic Mn-doped ZnO nanoparticles can be precipitated in the borate matrix if the glasses annealed at particular temperature.

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6. References

- [1] A. Pan and A. Ghosh, "A New Family of Lead-Bismuthate Glass with a Large Transmitting Window," *Journal of Non-Crystalline Solids*, Vol. 271, No. 1-2, 2000, pp. 157-161. [doi:10.1016/S0022-3093\(00\)00111-3](https://doi.org/10.1016/S0022-3093(00)00111-3)
- [2] Š. Jiri, K. Ladislav, M. Petr, M. Lionel, R. Bertrand and G. Ivan, "Structure and Properties of MoO₃-Containing Zinc Borophosphate Glasses," *Journal of Non-Crystalline Solids*, Vol. 355, No. 16-17, 2009, pp. 970-975. [doi:10.1016/j.jnoncrysol.2009.04.017](https://doi.org/10.1016/j.jnoncrysol.2009.04.017)
- [3] C. W. Adrian, "Borate Structures: Crystalline and Vitreous," *Physics and Chemistry of Glasses—European Journal of Glass Science and Technology Part B*, Vol. 51, No. 1, 2010, pp. 1-39.
- [4] M. Pal, "Structure and Physical Properties of Sodium Antimony Germinate Glasses," *Journal of Materials Research*, Vol. 11, No. 7, 1996, pp. 1831-1835. [doi:10.1557/JMR.1996.0231](https://doi.org/10.1557/JMR.1996.0231)
- [5] W. L. Konijnendijk and J. M. Stevels, "Structure of Borate and Borosilicate Glasses," In: L. D. Pye, V. D. Fréchet and N. J. Kreidl, Eds., *Borate Glasses: Structure, Properties, Applications*, Plenum Press, New York, 1978, p. 259.
- [6] I. Kashif, H. Farouk, A. S. Aly and A. M. Sanad, "Differential Scanning Calorimetry and Infrared Study of Barium Borate Glass Containing Transition Elements," *Physics and Chemistry of Glasses*, Vol. 32, No. 2, 1991, pp. 77-78.
- [7] A. C. Hannon, D. I. Grimley, R. A. Hulme, A. C. Wright and R. N. Sinclair, "Boroxol Groups in Vitreous Boron Oxide: New Evidence from Neutron Diffraction and Inelastic Neutron Scattering Studies," *Journal of Non-Crystalline Solids*, Vol. 177, No. 1, 1994, pp. 299-316. [doi:10.1016/0022-3093\(94\)90544-4](https://doi.org/10.1016/0022-3093(94)90544-4)
- [8] D. L. Griscom, "Borate Glass Structure," In: L. D. Pye, V. D. Fréchet and N. J. Kreidl, Eds., *Borate Glasses: Structure, Properties, Applications*, Plenum Press, New York, 1978, p. 11.
- [9] C. Li and Q. Su, "Action of Co-Dopant in Electron-Trapping Materials: The Case of Sm³⁺ in Mn²⁺ Activated Zinc Borosilicate Glasses," *Applied Physics Letters*, Vol. 85, No. 12, 2003, pp. 2190-2192. [doi:10.1063/1.1797562](https://doi.org/10.1063/1.1797562)
- [10] J.-M. Wu and H.-L. Huang, "Microwave Properties of Zinc, Barium, and Lead Borosilicate Glasses," *Journal of Non-Crystalline Solids*, Vol. 260, No. 1-2, 1999, pp. 116-124. [doi:10.1016/S0022-3093\(99\)00513-X](https://doi.org/10.1016/S0022-3093(99)00513-X)
- [11] L. D. Bogomolova and M. P. Glassova, "The Impurity Effects in Vanadate Semiconducting Glasses," *Journal of Non-Crystalline Solids*, Vol. 37, No. 3, 1980, pp. 423-426. [doi:10.1016/0022-3093\(80\)90079-4](https://doi.org/10.1016/0022-3093(80)90079-4)
- [12] M. Pal, D. Chakravorty and A. Bhowmik, "Structural Study of Iron Borate Glasses Containing NiO and ZnO," *Journal of Materials Research*, Vol. 13, No. 11, 1998, pp. 3287-3292. [doi:10.1557/JMR.1998.0447](https://doi.org/10.1557/JMR.1998.0447)
- [13] L. Aleksandrov, R. Iordanova and Y. Dimitriev, "Glass Formation in the MoO₃-La₂O₃-B₂O₃ System," *Physics and Chemistry of Glasses*, Vol. 48, 2007, p. 242.
- [14] R. M. Almedia and J. D. Mackenzie, "Vibrational Spectra and Structure of Fluorozirconate Glasses," *Journal of Chemical Physics*, Vol. 74, No. 11, 1981, pp. 5954-6537. [doi:10.1063/1.441033](https://doi.org/10.1063/1.441033)
- [15] S. Sakka and K. Kamiya, "Structure of Alkali Germanate Glasses Studied by Spectroscopic Techniques," *Journal of Non-Crystalline Solids*, Vol. 49, 1982, p. 103. [doi:10.1016/0022-3093\(82\)90110-7](https://doi.org/10.1016/0022-3093(82)90110-7)
- [16] P. Becker, "Thermal and Optical Properties of Glasses of the System Bi₂O₃ - B₂O₃," *Crystal Research and Technology*, Vol. 38, No. 1, 2003, pp. 74-82. [doi:10.1002/crat.200310009](https://doi.org/10.1002/crat.200310009)
- [17] A. H. Verhoef and H. W. den Hartog, "A Molecular Dynamics Study of B₂O₃ Glass Using Different Interaction Potentials," *Journal of Non-Crystalline Solids*, Vol. 146, 1992, pp. 267-278. [doi:10.1016/S0022-3093\(05\)80501-0](https://doi.org/10.1016/S0022-3093(05)80501-0)
- [18] S. Ram, "Infrared Study of the Dynamics of Boroxol Rings in the Crystallization of BaFe₁₂O₁₉ Microcrystal in the Borate Glass," *Physical Review B*, Vol. 51, No. 10, 1995, pp. 6280-6286. [doi:10.1103/PhysRevB.51.6280](https://doi.org/10.1103/PhysRevB.51.6280)
- [19] N. M. Bobkova and S. A. Khot'ko, "Zinc Oxide in Borate Glass-Forming Systems," *Glass and Ceramics*, Vol. 62, No. 5-6, 2005, pp. 167-170. [doi:10.1007/s10717-005-0064-7](https://doi.org/10.1007/s10717-005-0064-7)
- [20] W. Soppe, J. Kleerebezem and H. W. den Hartog, "Raman Spectroscopy Study of (B₂O₃)_{1-x-y}(Li₂O)_x(Li₂Cl₂)_y and (B₂O₃)_{1-x-y}(Li₂O)_x(Cs₂O)_y," *Journal of Non-Crystalline Solids*, Vol. 93, 1987, p. 142. [doi:10.1016/S0022-3093\(87\)80034-0](https://doi.org/10.1016/S0022-3093(87)80034-0)
- [21] L. Edwards, M. Gouterman and X. V. Porphyrins, "Vapor Absorption Spectra and Stability: Phthalocyanines," *Journal of Molecular Spectroscopy*, Vol. 33, No. 2, 1970, pp. 292-310. [doi:10.1016/0022-2852\(70\)90040-8](https://doi.org/10.1016/0022-2852(70)90040-8)
- [22] M. Pal, "Borate Based Spintronic Materials in Bulk Form above Room Temperature," *Journal of Surface Science and Technology*, Vol. 21, No. 1-2, 2005, pp. 91-96.